# Investigation on the properties of ambient cured alkali activated binder concrete

### M Talha Junaid<sup>1\*</sup>, Abdul Saboor Karzad<sup>1#</sup>, Moussa Leblouba<sup>1</sup>

### ABSTRACT

This research paper reports results of strength development properties of ambientcured alkali activated binders (AABs) and their reliance on different parameters, such as type of pre-cursor materials & their proportions, type & ratio of activators, and curing age. In this research study sodium silicate (SS) and sodium hydroxide (SH) solutions were used to activate a blend of low-calcium (class F) fly ash (FA) and ground granulated blast furnace slag (GGBFS). A total of 25 AABs mixes with various mix proportions are investigated in this research study. The results indicate that AABs can be manufactured with strength properties similar to ordinary Portland cement concrete (OPC). For further insight into the mechanical property variations of these samples SEM and EDS tests are conducted and their micrographs are presented here. A previously proposed analytical model to predict the compressive stress-strain response of these AABs is also modified in this study to better capture their post peak behaviour.

*Keywords:* Alkali activated binder, Elastic modulus, Compressive strength, Flexural strength, Stress-strain, SEM, EDS.

### **1. INTRODUCTION**

Concrete is a commonly used construction material for structure and infrastructure development and the most widely used material second only to water. Therefore, the yearly global production of cement being the main binder of concrete, will only increase due to high demand of infrastructure growth (Davidovits, 1994; US Geological-Survey, 2013; Shehab et al., 2016). The production of cement not only adds 7% of the total manmade greenhouse gas emission (CO<sub>2</sub>) to the atmosphere but also produces some other environmentally hazardous material such as SO<sub>3</sub> and NO<sub>x</sub> (Malhotra, 2002; McCaffrey, 2002; Fernández-jiménez, 2006; Lloyd and Rangan, 2010; Shehab et al., 2016). CO<sub>2</sub> emissions resulting from OPC production can be divided into two halves: one coming from calcination of limestone while the other one is due to the fossil fuel combustion for the kiln (Hardjito et al., 2008).

To tackle the increase of these environmentally harmful products due to production of Portland cement (PC), alternative pozzolanic materials like metakaolin, fly ash, silica fume, GGBFS, and rice husk ash have been used as cement replacement (Ganesh and Muthukannan, 2018). These pozzolanic materials have made some impact on the carbon footprint of cement, however, this has been rather limited. More recently, researchers have focused on development of binders that do not uses Portland cement in full (Hassan et al., 2019; Ahmed et al., 2021; Shrivas et al., 2021). Alkali activated binders (AABs) are one such group of binders that have come to the forefront as a possible replacement for OPC. AABs are formed by activating silica (SiO<sub>2</sub>) and alumina (Al<sub>2</sub>O<sub>3</sub>) rich industrial by-products using a strong metallic alkaline solution. (Davidovits and Orlinski, 1988; Davidovits, 1991; Lloyd and Rangan, 2010; Topark-Ngarm et al., 2015).



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AABs do not depend on calcination of limestone as is the case in Portland cement production. The calcination of limestone is one of the major contributors to the production of  $CO_2$  in Portland cement production which is known as man-made greenhouse gas emission (McCaffrey, 2002). Therefore, AABs are expected to reduce man-made greenhouse gas emissions by 80% (Kong et al., 2006; Fernández-jiménez, 2006; Lloyd and Rangan, 2010).

In the past two decades AABs as an alternative to OPC concrete, has been under focus and investigated with the view to industrialize this material. Adding to the advantage in reduction of CO<sub>2</sub> emissions, AABs can acquire similar or even superior mechanical properties to OPC concrete (Hassan et al., 2019; Chandrakanth and Koniki, 2020). Moreover, AABs exhibit high early strength, high durability and can maintain strength in severe conditions such as corrosive environment, elevated temperature, and high radiation (Binici, 2013). Despite the aforementioned advantages, there are many challenges facing AABs that need to be tackled in order to be used as a mainstream sustainable construction material. One of the main challenges is the reliance of the mechanical properties of AABs on the source of precursor materials and the activation system. The other challenge is varied setting time and low workability of high calcium fly ash AABs (Chindaprasirt et al., 2012; Topark-Ngarm et al., 2015). Finally, AABs require high temperature curing regime to attain substantial mechanical properties.

Several studies conducted on development and properties of AABs concluded that physical and mechanical properties of AABs is dependent on many variables: ratio of alkaline activator solution, SH concentration (Molarity), alkaline activator to fly ash ratio, amount of GGBFS as partial replacement to fly ash, and curing conditions (Fernándezjiménez, 2006; Diaz et al., 2010; Diaz-Loya, et al., 2011; Lee and Lee, 2013; Junaid et al., 2014a; Junaid et al., 2015, 2016; Junaid et al., 2017; Nath and Sarker, 2017). Many researchers reported on the general enhancement of AABs properties when fly ash is partially replaced by GGBFS (Ravikumar et al., 2010; Deb et al., 2014; Nath and Sarker, 2014).

Nath and Sarker (2014) added small quantities of GGBFS to low-calcium fly ash-based AABs that were cured at ambient conditions. They added GGBFS up to 30% of total binder and achieved a compressive strength of 55 MPa at 28 days. They also reported that concretes having silicate to hydroxide ratio equal to 2.5 showed decreased slump and setting time than the mixtures with 1.5 and 2.0 ratios. Nath and Sarker (2015) in another research, studied the relationship between the experimental and predicted values of modulus for AABs using a proposed analytical model. They reported that the modulus of ambient cured AABs is 25-30% lower than that of OPC.

Another study by Deb et al. (2014) found good correlation between the development of the tensile strength and the compressive strength of ambient cured fly-ash AABs that were blended with slag. It was concluded that the methods of predicting the tensile strength of OPC concrete from their compressive strength recommended in AS 3600 and ACI 318 codes worked well with the ambient cured AABs. Hardjito et al. (2008) studied the setting time and strength properties of low calcium fly ash AABs and concluded that the compressive strength increases as the concentration of alkaline activator increases. A study conducted by Jaarsveld et al. (Van Jaarsveld et al., 2002) investigated the effect of curing process on properties of AABs, found that curing at elevated temperature resulted in cracking and can impose adverse effect on properties of GPC

Despite these investigations on properties and behaviour of AABs, there is a need for extensive investigation regarding mechanical particularly properties (i.e., compressive strength, modulus of elasticity, flexural strength, etc.) of ambient cured AABs. On the other hand, for AABs to become a dependable building material a reliable stress-strain relationship needs to be established. The focus of this experimental program is to study the mechanical properties of ambient cured AABs. To this end, 25 different types of mixes were prepared and around 400 samples were tested. The chemical composition and morphology of these ambient cured AABs were also studied. Finally, a stress-strain model is presented that accurately predicts the compressive response of such concretes.

#### 2. EXPERIMENTAL PROGRAMME

#### 2.1 Materials

Locally attained GGBFS and low calcium fly ash (Class F) imported from India was used as pre-cursor material in this study. Table 1 shows chemical composition of the fly ash and GGBFS used in this experimental program as found using X-Ray Fluorescence (XRF). The precursor material was activated using a combination of Sodium silicate (SS) and sodium hydroxide (SH) solutions. SS solution with density of 1.296 to 1.396 g/ml (produced by German based company Merck KGaA and procured from Emirates Scientific & Technical Supplies (ESTS), UAE) was used. The SS solution had 27% SiO and 8% Na<sub>2</sub>O. SH solution (12M) was made by dissolving 98% purity flakes in water. Only one molarity of SH solution was used in this study. Coarse aggregates with a nominal size of 10 mm, absorption of 0.7%, and specific gravity of 2.68, were used, while fine aggregates were a blend of 5 mm crushed aggregates and dune sand, in the ratio of 60-40 percentage. The specific gravity of these combined fine aggregates was 2.62 and an absorption of 1%.

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_	Table 1. Chemical composition of my ash and GODT 5 using x ray hubbescence (XRC)									)				
	Material	$Al_2O_3$	$SiO_2$	CaO	$SO_3$	Na <sub>2</sub> O	MgO	K <sub>2</sub> O	TiO <sub>2</sub>	$MnO_2$	Fe <sub>2</sub> O <sub>3</sub>	NiO	ZnO	SrO
	GGBFS	5.7	32.4	50.6	2.4	-	6.5	-	0.5	0.7	0.6	-	-	-
	FA	10.9	59.8	8.7	1.5	1.0	7.5	1.4	1.1	0.2	7.5	0.03	0.04	0.08

Table 1. Chemical c	omposition	of fly ash and	GGBFS using x-ra	y fluorescence (	(XRF)
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#### 2.2 Specimens Preparation

The mixing procedure for making AABs is similar to that of the OPC concrete. Alkaline solutions were prepared at least 24 hours before mixing. The mixing was done at room temperature. Fine and coarse aggregates in saturated surface dry (SSD) condition were mixed with fly ash and GGBFS for 2 minutes with some portion of water. Silicate's solution, if present in the mix, was then added and mixed for 3 minutes. Hydroxide was then added and mixed for another 2 minutes. The required amount of water was then added and mixed for another 2 minutes. The framework provided by Junaid et al. (2015) was used to proportion the mixes. The water to binder ratio was calculated using the following equation:

$$w/b = \frac{W_{free} + W_{OH} + W_{Si}}{FA + G + S_{OH} + S_{Si}}$$
(1)  
Where;

 $W_{free}$  = Weight of free water in the system  $W_{OH}$  = Weight of water content of the OH solution  $W_{Si}$  = Weight of water content of the silicate solution FA = Weight of Fly Ash G = Weight of GGBFS

 $S_{OH}$  = Weight of solid content of the OH solution

 $S_{Si}$  = Weight of solid content of the silicate solution

A total of five Sets, each consisting of five different mixes (resulting in a total of 25 different mixes) were cast in this study. The only variable within each Set was the ratio of GGBFS to fly ash (i.e., from zero% in mix1 to 100% in mix5-except Set 4 and 5) while all the other proportions were kept constant throughout the mixes in each Set. Between Sets, the Sodium Silicate to Sodium Hydroxide ratio was changed to study its effect on the mechanical properties. The Geopolymer solids to Alkali Liquid (GS/AL) ratio was kept constant for all sets in the study. However, depending on the workability requirements additional water was added to the mix. The differences between the sets are the ratio of SS to SH (denoted by letter R) and the w/b ratio. Table 2 summarizes the mix design proportions of the mixes used in this study.

Cylinders and prisms with nominal dimensions of 100x200 mm and 100x100x500 mm respectively were cast. The cylinders were filled layer by layer and compacted by manual strokes using a steel tamper. Additionally, after the

samples were filled, they were further compacted using the vibration table. The exposed surfaces of the specimens were wrapped in watertight membrane after casting to minimize water loss during setting. After 24 hours the sample were removed from their moulds and kept in room temperature until time of testing.

#### 2.3 Test Scope

Compressive strength, splitting tensile strength and modulus of elasticity tests were performed on cylindrical specimen according to the ASTM standard tests ASTM C39/C39M - 16b (2016), ASTM C496 (2006), and ASTM C469/C469M (2013), respectively. The flexural strength tests were performed on prismatic beams according to the ASTM C78/C78M (2002). The compression test was conducted using a 3000 kN hydraulic compression machine (shown in Fig. 1(a)) with a pace rate of 2.4 kN/s, while splitting or indirect tensile testing test was conducted at a slower pace rate of 0.94 kN/s using the same machine.

Furthermore, the stress-strain response of the samples was captured by testing the cylindrical samples at 28 days under compression in deformation control mode using rings as per the standard. LVDTs mounted on rings were placed on either side of the specimen to capture the deformation. In the current study, for each mix one cylinder was tested with the rate of 0.01 mm/sec until the load level reached half the expected strength of the sample. Afterwards, as programmed the loading rate automatically changed to 0.005 mm/sec until the failure of the sample.

Scanning electron microscopy (SEM) and Energy Dispersive X-ray Spectroscopy (EDS) testing was conducted on samples particles collected during destructive tests (Cylinders and prisms tests) in order to record the morphology and analyse the reaction products. SEM test was performed by taking images using Tescan VEGA 3 XMU while the EDS spectra test was performed using Oxford Instruments X-max 50 EDS and an SDD and LN free detector with a resolution of 127eV. The samples were kept under desiccator for drying. A thin layer of Carbon was deposited using Quorum Technology carbon evaporator prior to the SEM analysis. All SEM tests were done at 28 days.

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				Pre-cursor		Activators			Aggregates			w/b	Unit volume	
	Mix ID		materials kg/m <sup>3</sup>		kg/	kg/m <sup>3</sup>			kg/m	kg/m <sup>3</sup>		W/ U		
					വ	55	D	Coarse		Fine	ka/m <sup>3</sup>	Datia	m <sup>3</sup>	
			ĨA	UODI'S	511	55	K -	10 mm	5 mm	Dune Sand	Kg/III	Katio	111	
		S1M1	425.0	0.0	165.0	0.0	0.0	1065	390	260	50.0	0.32	1.0	
	Set1	S1M2	340.0	85.0	165.0	0.0	0.0	1065	390	260	50.0	0.32	1.0	
		S1M3	212.5	212.5	165.0	0.0	0.0	1065	390	260	50.0	0.32	1.0	
		S1M4	85.0	340.0	165.0	0.0	0.0	1065	390	260	50.0	0.32	1.0	
		S1M5	0.0	425.0	165.0	0.0	0.0	1065	390	260	50.0	0.32	1.0	
		S2M1	425.0	0.0	82.5	82.5	1.0	1065	390	260	58.5	0.32	1.0	
		S2M2	340.0	85.0	82.5	82.5	1.0	1065	390	260	58.5	0.32	1.0	
	Set2	S2M3	212.5	212.5	82.5	82.5	1.0	1065	390	260	58.5	0.32	1.0	
		S2M4	85.0	340.0	82.5	82.5	1.0	1065	390	260	58.5	0.32	1.0	
		S2M5	0.0	425.0	82.5	82.5	1.0	1065	390	260	58.5	0.32	1.0	
	Set3	S3M1	425.0	0.0	55.0	110.0	2.0	1065	390	260	60.0	0.32	1.0	
		S3M2	340.0	85.0	55.0	110.0	2.0	1065	390	260	60.0	0.32	1.0	
		S3M3	212.5	212.5	55.0	110.0	2.0	1065	390	260	60.0	0.32	1.0	
		S3M4	85.0	340.0	55.0	110.0	2.0	1065	390	260	60.0	0.32	1.0	
		S3M5	0.0	425.0	55.0	110.0	2.0	1065	390	260	60.0	0.32	1.0	
	Set4	S4M1	425.0	0.0	47.0	118.0	2.5	1065	390	260	60.0	0.32	1.0	
		S4M2	403.7	21.3	47.0	118.0	2.5	1065	390	260	60.0	0.32	1.0	
		S4M3	382.5	42.5	47.0	118.0	2.5	1065	390	260	60.0	0.32	1.0	
		S4M4	361.3	63.7	47.0	118.0	2.5	1065	390	260	60.0	0.32	1.0	
		S4M5	340.0	85.0	47.0	118.0	2.5	1065	390	260	60.0	0.32	1.0	
		S5M1	425.0	0.0	55.0	110.0	2.0	1065	390	260	30.0	0.26	1.0	
	Set5	S5M2	382.5	42.5	55.0	110.0	2.0	1065	390	260	30.0	0.26	1.0	
		S5M3	340.0	85.0	55.0	110.0	2.0	1065	390	260	30.0	0.26	1.0	
		S5M4	297.5	127.5	55.0	110.0	2.0	1065	390	260	30.0	0.26	1.0	
		S5M5	255.0	170.0	55.0	110.0	2.0	1065	390	260	30.0	0.26	1.0	

Table 2. Mix design of the AAB concrete studied in this research paper

#### **3. RESULT AND DISCUSSIONS**

#### 3.1 Strength Properties

#### 3.1.1 Compressive and Splitting Tensile Strength

The 7 and 28-days average compressive and splitting tensile strength of the cylinders are illustrated in Fig. 1. From each mix, three cylinders were tested at 7 and 28 days, and the average test value is reported here.

Fig. 1(b) shows the strength development of samples in Set 1 with varying percentages of GGBFS. It is worth nothing that this set only used hydroxide as the sole activator (R = 0). The maximum compressive strength at 7 days reached for this Set was 16 MPa for samples of S1M5 with 100% GGBFS. All other mixes also returned similar strength values with the only exception being Mix 1 that showed a very low compressive strength (approximately 3 MPa). Although addition of GGBFS improved the compressive strength of this set, the increase was rather small suggesting that single activator results in weak binders unfit for structural applications.

In contrast to Set 1, the compressive strength development showed a very different trend for mixes in Set 2 (R = 1). This set showed a high dependency on the amount of GGBFS used in the mix. Mix 1 (S2M1) with only fly ash

at 7 days. However, by adding GGBFS a significant increase in the compressive strength was observed. Fig. 1(c) shows the highest compressive strength for this Set was around 42 MPa for mixes containing more than 50% GGBFS. A strength increases from 9.3 MPa to 23.6 MPa was observed when the GGBFS amount was changed from 20 to 50 percent for this set. When the GGBFS amount was further increased from 50 to 80 percent, the strength values changed from 23.6 MPa to 41.5 MPa. The change in strength for these mixes (i.e. from 9.3 MPa to 23.6 MPa and from 23.6 MPa to 41.5 MPa) that shows an increase of 2.5 times and 1.6 times, respectively; were found to be in the same proportion as the change in the amount of GGBFS (from 20% to 50% and from 50 to 80%). Any further increase in the amount of GGBFS (100%) to the mixes did not change the compressive strength of the specimen. Since for S2M1 and S2M2, the observed compressive strength is below 20MPa at 7 days, these mixes may not be appropriate for structural applications. However, as the percentage of GGBFS is increased beyond 20% the resulting binder showed good mechanical strength properties.

as precursor material had a negligible compressive strength

For Set 3 (R = 2.0) the pattern of strength development with respect to the percentage of GGBFS added to the mix was almost similar to the Set 2 strength development pattern



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as shown in Fig. 1(d). Mix 4 and 5 in Set 3 reached a compressive strength of around 50 MPa at 7 days. As observed in Set 2, these mixes gained substantial strength when the GGBFS percentage was 50 and 80%. Any further increase in the GGBFS amount did not have a positive effect on the strength development of the mixes. Similar to Set 2, mixes 1 and 2 are not suitable for strucutal application due to their low compressive strength.

Set 4 was designed to have low proportions of GGBFS, with the maximum of 20% replacement. These mixes also had a high silicate to hydroxide ratio (R = 2.5). It was found that the strength development of these mixes was immune to the addition of GGBFS up to 10%. As the GGBFS percentage was increased beyond 10% a slight jump in the strength development was observed. Increasing the GGBFS amount any further resulted in no change of compressive strength. However, the strength development of these mixes was not appropriate for use in structural engineering applications. As can be seen in Fig. 1(f), for Set 5 the gain in strength is proportional to the increase in GGBFS in the system. The maximum strength for this set was around 50MPa at 7 days. Also of note is the fact that for the mix S5M5 the maximum amount of GGBFS added was 40% of the total binder.

Further discussion on the strength development from a morphology point of view is made later in this paper (section 3.3 and 3.4).

As expected, the 28-day strength of all tested samples exhibited a similar trend to the 7-day strength as depicted in Fig. 1. Generally speaking, the compressive strength in all tested samples did not change significantly after the first 7 days period. This is also in agreement with earlier research (Li et al., 2017; Wattimena et al., 2017) which reported that unlike OPC the compressive strength of AABs develops in the first 7 days. After this initial period, there is not significant change in the strength of such concretes. However, current results obtained from Set 1 show a noticeable gain in strength after the initial 7 days period. This may indicate a dependency of strength development on the type of activator used. Other factors that may affect this behaviour are the precursor type and their proportions, although more work is required to further investigate this effect and draw a definite conclusion. From the compressive strength results shown in Fig. 1 and Fig. 2 a, it can be concluded that the addition of higher amounts of GGBFS and sodium silicate resulted in higher overall compressive strength.

The evolution of the splitting strength of individual mixes is shown in Fig. 1 (b-f on the axis on the right) while Fig. 2(b) illustrates the development of splitting strength for all sets. It can be seen from Fig.s 1 (b-f) that there is no change in the splitting strength of AABs with time. It can therefore be concluded that AABs attain all their splitting tensile strength in the first 7 days after casting. This may be a desirable property to minimize the early cracking potential of AABs. Moreover, similar to OPC, the splitting tensile strength of AABs tested in this program return a value of almost 7-10% of their corresponding compressive strength. The maximum and minimum tensile strength reported were 3.9 MPa and 0.2 MPa for S5M5 and S2M1, respectively.

#### 3.1.2 Flexural Strength

The flexural strength of specimens can be used as an indication of the tensile strength of concrete. Despite that, the flexural strength is, in general, higher than the indirect splitting tensile strength. Fig.3 (a) shows the flexural strength at 28 days of all 5 sets obtained by testing the standard beam samples  $(500 \times 100 \times 100 \text{ mm})$  in four points loading setup as shown in Fig. 3(b). In general, and as expected, the flexural strength test results show very similar pattern to the tensile and compressive strength of the mixes studied in this paper.

It is clear from Fig. 3 that the flexural strength increased with the increase in the percentage replacement of fly ash with GGBFS, in all sets except in Set 1 (which contains only SH as alkali activator), where after mix 3 the flexural strength started to deplete. The highest flexural strength was achieved by Set 2 and 3 with the AABs mix having 100% GGBFS content.

It is also observed from Fig. 3 that increasing the alkali solution ratio (sodium silicate / sodium hydroxide) has a





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Fig. 3. (a) Flexural strength at 28 days for all 5 sets, (b) Flexural strength test setup



Fig. 4. (a) Results of young's modulus test at 28 days, (b) Young's modulus test setup

positive impact in improving the flexural strength. However, this is true up until a certain ratio (R). In particular, when R increased from 2.0 (Set 5) to 2.5 (Set 4), the flexural strength values decreased significantly. According to Nath and Sarker (2015), flexural strength for mixtures having 40% alkali activator by volume of total mix showed increasing trend with the replacement of fly ash by GGBFS until 10% and decreased after that. This implies that the amount and ratio of the activating system affect the flexural strength of AAB specimens until certain values. Further investigations of a combined effect of these two variables may be of concern for the researchers in the near future.

#### 3.1.3 Elastic Modulus and Density

Compared to OPC, limited data is available on moduli values of AABs. However, it is widely accepted that the

modulus of elasticity of AABs is generally lower than OPC samples of similar strengths (Nath and Sarker, 2017). This reduction in the moduli will dictate the flexural and axial stiffness of members made using AABs. It is therefore of importance that modulus of elasticity for ambient cured AABs be determined. Fig. 4 shows the experimental setup and the 28-day modulus of elasticity for specimens in this study.

The highest modulus values were consistently reported for Set 1 and were in good agreement with prediction models recommended by EuroCode 2 and ACI 318-14. Interestingly, the modulus values for Set 2 were significantly lower compared to Set1, although samples in Set 2 returned a higher compressive strength. The evolution of elastic modulus for Set 3 was very similar to the trend exhibited by its compressive strength, reaching a maximum value of 20 GPa. A similar trend was observed for Sets 4 and

Mechanical	Compressi	ve strength	Split tens	ile strength	Flexural	Elastic modulus	Density		
property	(M	Pa)	. (N	IPa)	strength (MPa)	(E) (GPa)	(kg	$(kg/m^3)$	
Mix ID	Mix ID 7 Days 28 Days		7 Days	28 Days	28 Days	28 Days	7 Days	28 Days	
S1M1	2.84	12.1	1.2	1.7	2.81	17.3	2318	2290	
S1M2	12	17.3	1.82	2.14	2.86	22.1	2389	2319	
S1M3	14.1	19.1	2.48	2.45	3.5	20.3	2317	2308	
S1M4	15.8	22.4	2.35	2.77	2.4	25	2344	2334	
S1M5	16	22	2.24	2.78	2.6	25.4	2306	2299	
S2M1	0	1.45	0.64	0.24	0.2	5.8	2286	2230	
S2M2	9.3	11.96	1.53	1.26	2.6	5.2	2315	2318	
S2M3	23.6	24.9	2.67	2.55	2.5	12.2	2335	2356	
S2M4	41.5	45.8	3.75	3.44	3	24.4	2313	2359	
S2M5	42.2	42.5	3.25	3.5	5	20.5	2337	2375	
S3M1	3.5	4	0.5	0.6	0.75	6.2	2299	2230	
S3M2	11.5	8.5	1.4	1.1	2.2	10.3	2335	2307	
S3M3	37.6	42.1	2.6	2.2	2.6	14.5	2410	2398	
S3M4	51	55.9	2.8	3.2	4.15	20.4	2406	2339	
S3M5	50.7	55	3.6	3.3	5	17.9	2395	2342	
S4M1	2.65	3.1	0.4	0.55	0.47	9.1	2239	2259	
S4M2	3.5	4.6	0.535	0.825	1.45	11.8	2282	2268	
S4M3	3.85	3.85	0.525	0.725	1.35	12.5	2277	2244	
S4M4	15.1	16	1.45	1.3	3.24	11.9	2330	2244	
S4M5	15	16	1.6	1.35	2.75	13.5	2386	2308	
S5M1	18	21	1.6	2	1.22	13.8	2442	2413	
S5M2	29	27	2.6	2.7	2.12	18.9	2422	2397	
S5M3	35.5	37	2.9	3	2.34	22.5	2437	2448	
S5M4	38	43	2.8	3.2	3	21.0	2458	2410	
S5M5	49	55	3.8	3.9	3	24.0	2454	2426	

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5, with maximum values of 14 GPa and 24 GPa respectively. It may therefore be concluded that the addition of GGBFS increases the elastic modulus of ambient cured AABs, while the inclusion of sodium silicate reduces this property (Lee and Lee, 2013). It may also be concluded that the modulus

values of ambient cured AABs is a function of its compressive strength, however, it does not necessarily follow the same prediction equations as established for OPC concretes. Tabulated summary of the results reported above is presented in Table 3. Fig. 5 shows the average density of all 5 sets which is calculated by dividing weight of cylindrical samples to its volume. The volume was determined using the average length and diameter of the samples. The average density of all samples was around 2350kg/m<sup>3</sup>, which is comparable to ordinary concrete (Mindess et al., 2003; Neville and Brooks, 2010).

#### 3.2 Stress – Strain Curves

The behaviour and material properties of AABs may differ from OPC concrete due to the difference in their composition. Therefore, it is important for the purposes of design and simulation to understand the stress-strain behaviour of AABs. However, unlike the conventional OPC concrete that has a relatively well-known stress-strain behaviour, the stress-strain of geopolymer concrete is still not fully understood (Yadollahi and Benli, 2017) [36]. Therefore, the stress-strain behaviour of AABs prepared in this study was determined at 28 days and is presented in Fig. 6. As shown in the Fig.6 (a), Set 1 shows a strain of 0.002 at maximum stress, which continued to increase until full failure of the sample. This is also typical behaviour for OPC samples and design recommendations use this strain value for maximum stress.

However, the mixes in the remaining four sets that have both the activators (SH and SS) exhibited a strain value higher than the OPC concrete. Mix 4 and mix 5 of Set 2 (S2M4 and S2M5) exhibited a strain value of 0.003 at the maximum stress. However, the first three mixes of Set 2 that did not have structural application strength, strained higher than the strain limit of OPC concrete. Similarly, Set 3 showed strain values higher than OCP concrete for the mixes that have compressive strength in excess of 30 MPa, and even higher strain for the mixes which have lower compressive strength. On the other hand, Set 4 exhibited high strain values for only mixes 4 and 5 that have high compressive strengths, and very low strains as well as low compressive strength for the other three mixes in the set. Set5 mixes which all have high compressive strengths, exhibited strain values higher than OPC concrete at the peak load. It can be inferred from Fig. 6 that the peak point position is highly influenced not only by the strength of the sample but also by the amount of GGBFS added to the mix and the alkali activating system.

As presented in Fig. 6, all the samples in Set 1 (which were activated by SH solution only) provided good stress strain behaviour that is comparable to conventional OPC concrete. Whereas in Sets 2 and 3, mixes 4 and 5 (with higher percentages of GGBFS) showed more brittle failure compared to the rest of the mixes, with a sudden decline in stress in the post-peak softening stage.

The results of the stress-strain test conducted in this study were compared to the analytical model proposed by Junaid et al. (2017) as shown below. The model considered, which is derived from Collins et al. (1993), has the following form:  $\sigma_c = f'c \cdot \frac{\varepsilon_c}{\epsilon_{cm}} \cdot \frac{n}{n-1+(\varepsilon_c/\epsilon_{cm})^{nk}} \qquad (MPa) \qquad (2)$ 

Where  $\varepsilon_c$  is strain at stress  $\sigma_c$ , f'c is the peak compressive strength, and  $\varepsilon_{cm}$  is the strain at maximum strength. n and k factors are unitless and are found as following:

$$n = 0.7 + (f_c'/23) \tag{3}$$

$$k = 0.6 + (f_c / 86) \text{ when } \varepsilon_c / \varepsilon_{cm} > 1,$$
  
and  $k = 1.0 \text{ when } \varepsilon_c / \varepsilon_{cm} \le l$  (4)

For each value of  $\varepsilon_c$ , the values of  $\sigma_c$  is computed using Equation (2), with factors *n* and *k* calculated using equations 3 and 4. However, this model does not account for the post peak behaviour of such concretes. A post peak reduction factor ( $\lambda$ ) was therefore proposed in the current study and used for the considered analytical model. To capture the post peak behaviour,  $\lambda$  is multiplied to the *k* value obtained from Equation (4), only after peak stress is reached. The post peak reduction factor may be calculated using the following equations:

For 
$$f'c \le 25$$
,  $\lambda = 1.4 \times \frac{25}{\varepsilon_{cm}} \times 10^{-4} \le 2$  (5)

For 
$$f'c > 25$$
,  $\lambda = 1.1 \times \frac{f'c}{\varepsilon_{cm}} \times 10^{-4} \le 2$  (6)

Fig. 7 presents the results of the experimental stress-strain data along with the predictions by Junaid et al. (2017) model including the post peak reduction factor as proposed in this study. It can be noticed in Fig. 7 that the stress-strain behaviour of all the selected specimens is in good agreement with the models' predictions. Therefore, it can be concluded that Junaid et al. (2017) analytical model can be used to predict the stress-strain behaviour of AAB specimens.





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Fig. 8. SEM images for: (a) S1M1, (b) S1M2, (c) S1M4, (d) S1M5

#### 3.3 Scanning Electron Microscopy (SEM)

Scanning electron microscopy (SEM) was conducted on all the mixes of this study in order to understand the morphology of the different specimens and analyse their reaction products. The correlation between the microstructure of the different mixes and their mechanical behaviour can also be understood using this test. In Set 1 (which was activated using only sodium hydroxide) the reaction gel detected was less than the rest of the mixes which contain the additional sodium silicate solution. This was more pronounced when only fly ash was used as a reactive solid. It was also noticed that the increase in GGBFS content in the AAB mix not only increases the amount of reaction gel, but also produces more compact and homogeneous specimens. In the case of S1M1 and S1M2 the SEM graphs show a heterogeneous structure with

unreacted fly ash particles, and other regions of discontinuity in the geopolymer paste as seen in Fig. 8 (a) and (b).

The presence of unreacted fly ash particles indicates that the degree of reaction is low to moderate. On the other hand, Fig. 8 (c) and (d), for the mixes containing 80% and 100% GGBFS respectively, show more homogeneous, compact, and uniform microstructure, with no detection of individual unreacted or partially reacted particles. Fly ash particles usually need more time to be fully react compared to GGBFS particles due to larger surface area (Soutsos et al., 2016).

Fig. 9(a) and (b) shows that when a higher amount of GGBFS was added in the AAB mix, the reaction gel was found to be more homogeneously distributed in the paste, rather than acting as a shell surrounding the particle as in

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Fig. 9. SEM images for: (a) S3M1, (b) S3M5

the case where fly ash was the dominant binder. Fig. 9(a) shows fly ash particles with very thin layer of the geopolymer reaction gel. This thin layer may inhibit the production of a further reaction gel (Xie and Kayali, 2014) due to preventing more alumina and silica to dissolute from the surface of the fly ash particle.

As shown in Fig. 10 (a) the micrograph of Mix S5M4 shows a very compact specimen with no detectable microcracks, no unreacted particles, and with reaction gel present everywhere in the sample. More gel formation in the AABs matrix results in a higher compressive strength (Xie and Kayali, 2014). This was the reason for the high compressive strength for mix S5M4. Furthermore, the micrograph for mix S2M1 that recorded very low compressive strength, had a porous microstructure with large voids as shown in Fig. 10 (b).

It is also worth mentioning that the micrograph of the specimens of the different sets in this study indicated that an increase in the ratio of sodium silicate to sodium hydroxide (SS/SH) resulted in a smoother and denser paste with less pores and less fractured surfaces. Fig. 11(a) to Fig. 11(e) show how the increase of silicate solution decreased the number of unreacted fly ash particles and reduced the number of micro-cracks while GGBFS is absent.



Fig. 10. SEM images (a) S5M4, (b) S2M1



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Fig. 12. EDS result of mix S1M1

3.4 Energy Dispersive X-ray Spectroscopy (EDS)

Energy Dispersive X-ray Spectroscopy (EDS) was also conducted to further investigate the effect of composition on the mechanical properties of such concretes. In fly ash based AABs, N-A-S-H and C-A-S-H are considered the main reaction products of the geopolymerisation. Whereas when GGBFS is added to the mix, C-S-H gel becomes the main reaction product (Nath and Sarker, 2014; Xie and Kayali, 2014). The Si to Al ratio for the reaction product measured from the quantitative EDS of mix S1M1 (Fig. 12) was found to be 2.0, which falls within the range (1.8-2.0) of reported values for the alumino-silicate gel resulting from activating fly ash by only SH as the alkali activator (Junaid et al., 2014b). Meanwhile, Na/Al ratio of 1.5 indicated higher amount of reaction product, mainly N-A-S-H gel. Unreacted fly ash particles had lower amounts of Na, and high amounts of Si, Al, and O as predicted. Moreover, some traces of calcium carbonates (CaCO<sub>3</sub>) were detected in some areas, mainly the areas with white-colour material in the SEM micrograph of Fig. 12. This can be further corroborated by studying the EDS images where Ca, O, and C overlap.

Lower Al amount was found in S1M2 mix sample due to the inclusion of GGBFS which contains lower amounts of alumina than fly ash, and higher Ca/Si ratio (0.72) indicated the presence of C-S-H gel in this mix. In mix S1M3, a relatively large amount of Zn, Fe, K, and Mg besides Si, Na, Al, and Ca were detected as shown in Fig. 13. According to Duxson et al. (2007), polymerization is influenced by impurities like haematite that fill the actual polymeric binders and as a result affect the strength development of the concrete matrix to a great extent. C-A-S-H was found to be the main reaction gel in this mix due to high ratios of Si/Al and Ca/Si, along with rich amounts of N-A-S-H and C-S-H gels due to high ratios of Na/Al and Ca/Si, respectively. The coexistence of N-A-S-H and C-A-S-H was also reported in literature (Deb et al., 2014). The presence of white crystals with high Fe content, most probably ferrous or ferric oxides, was noticed in the mixes with 80 and 100 percent FA.

The mixes of set 2 and 3 generally had high amounts of all three reaction gels, with C-S-H gel being the dominant. Very high accumulation of calcium carbonate in some areas with some Ca, C, and O was also reported in these two sets, especially with higher replacement of fly ash by GGBFS. In addition to that, noticeable quantities of impurities such as K, Ti, S, and Fe existed almost in all areas of the samples of sets 2 and 3.

Sets 4 and 5 (having sodium silicate to sodium hydroxide ratios of 2.5 and 2.0, respectively) experienced little to moderate quantities of reaction products (dominant was C-A-S-H) that were poorly distributed in the concrete samples, and in some cases only one type of reaction gel was detected. It is shown in Fig. 14 and Fig. 15 that the samples in this set were filled with many unreacted particles of highly concentrated amounts of Si, Al, and O. This supports the findings of lower compressive strength recorded by the samples of this set as mentioned earlier in this paper. This could be attributed to the lower Na<sub>2</sub>O/SiO<sub>2</sub> molar ratio in the AABs mix when varying the sodium silicate to sodium hydroxide ratio in the alkali solution from 1.5 to 2.5 which leads to slight drop in the overall strength as also reported by Nath and Sarker (2015).



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Fig. 14. EDS result of mix S4M1





Fig. 15. EDS result of mix S4M4

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#### 4. SUMMARY AND CONCLUSIONS

This paper presents results from an experimental study conducted to determine the mechanical and morphological properties of ambient cured alkaline activated binder (AABs). The testing program determines the dependency of AABs' strength characteristics on various parameters such as proportion and type of precursor, ratio and type of activators used, and curing time. The strength properties reported are the compressive strength, splitting tensile strength, flexural strength and modulus of elasticity. Moreover, the compressive stress-strain response of AABs is also determined. Based on the findings of this study the following conclusions can be drawn:

- Adding sodium silicate (SS) solution to the AAB concrete can increase the strength of the binder however it results in low modulus of elasticity of the produced concrete.
- The gain in compressive strength from 7 days to 28 days is relatively slow in the samples activated by both sodium hydroxide (SH) and sodium silicate (SS), when compared to the samples of the mixes activated by SH only.
- The compressive strength and splitting tensile strength of AAB concrete follow a similar trend in strength development. Similar to OPC, the splitting tensile strength values were approximately 7-10% of their corresponding compressive strength. Moreover, there was no change in splitting tensile strength after the first 7 days.
- Flexural strength is also a fraction of the compressive strength value and follows a trend similar to its compressive strength. Moreover, increasing the amount of SS in the mix has a positive impact on the flexural strength up until a certain ratio of SS to SH.
- The elastic modulus of AAB concrete is usually lower than OPC of comparable strength. However, the addition of GGBFS increases the elastic modulus, in contrast with the addition of SS, which reduces the elastic modulus of the AAB concrete samples. The density of AAB concrete is comparable to that of OPC concrete.
- Stress-strain curves behaviour of the samples in this study indicates that for the ratio of SS to SH up to 2, a higher percentage replacement of fly ash by GGBFS gives a higher compressive strength but a lower peak strain. In addition, increased GGBFS and SS amounts added to the AABs concrete system gives a more brittle failure. The stress strain behaviour of AABs can be predicted by established analytical models as shown in the results.
- SEM analysis results supports the strength tests results by showing positive effect of increasing the amount of GGBFS added to the mix with the increase of (SS/SH) ratio, up to a ratio of 2. Beyond this, more unreacted

particles and micro-cracks in the paste were detected leading to decrease in strength.

• EDS images show high accumulations of all three types of reaction gels (C-H-S, N-A-S-H and C-A-S-H) in the mixes that recorded high strengths, along with noticeable quantities of chemical impurities that improves the strength of the AAB concrete samples.

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